

Scaling of the Anomalous Hall Effect in Ferrimagnetic Co₉₀Gd₁₀ Thin Films

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We investigated the anomalous Hall effect and longitudinal magnetoresistance in ferrimagnetic Co₉₀Gd₁₀ thin films with thicknesses from 5 to 35 nm and temperatures from 80 to 300 K. As the temperature increases from 80 to 300 K, the saturation magnetization of Co₉₀Gd₁₀ increases from 720 to 844 emu/cm³. However, the scaling law $\sigma_{xy} - \sigma_{xx}^n$ with $n = 2.3$ holds surprisingly well. Our results indicate that the scaling law usually reported in ferromagnetic materials remains valid for ferrimagnetic materials, which undoubtedly suggested the universality of the scaling law.

Index Terms—Anomalous Hall effect (AHE), ferrimagnet, magnetoresistance, scaling law.

I. INTRODUCTION

THE anomalous Hall effect (AHE) in ferromagnetic materials has attracted much attention for more than a century due to its physical complexity and important applications [1]–[10]. It manifests that electrons will acquire a spin-dependent transverse velocity when moving through a magnetized ferromagnetic conductor. For different spin orientations, electrons are deflected toward their opposite directions. Thus, the nonzero spin polarization in ferromagnetic materials directly leads to a net transverse Hall voltage. In general, in ferromagnetic materials, the anomalous Hall resistivity ρ_{xy} strongly depends on the material's magnetization. They are usually described in the empirical relation as [1]

$$\rho_{xy} = \mu_0 R_0 H + \mu_0 R_S M \quad (1)$$

where μ_0 is the vacuum magnetic permeability, H is the applied magnetic field, M is the magnetization, and R_o and R_S are the ordinary and anomalous Hall coefficients, respectively. The first term on the right side of (1) describes the ordinary Hall effect, which is caused by the Lorentz force acting on the moving charged carriers. The second term represents the AHE, usually much larger than the first one [11].

Although AHE of ferromagnetic materials has been extensively studied, the origin of AHE has still been an intriguing but controversial issue. Nowadays, this effect has been explained by three typical mechanisms: 1) intrinsic contributions [3], [12], [13]; 2) skew scattering [14]; and 3) side jump [15]. In experiment, the detailed relation between ρ_{xy} and the longitudinal resistivity ρ_{xx} are dominated by their underlying mechanisms [16]–[19]. The intrinsic contributions, proposed by Karplus and Luttinger, is considered as a purely quantum-mechanical origin of AHE [3]. They pointed out theoretically that AHE was related to the spin-orbit coupling in Bloch bands in a

perfect ferromagnetic crystal. Recently, such an intrinsic contribution has been expressed by a momentum-space Berry phase linked to the electronic structure of the multiband spin-orbit coupled system [20]. For this intrinsic contributions, $\rho_{xy} - \rho_{xx}^2$ is followed. The skew scattering is proposed by Smit [14]. In this theory, the spin-orbit coupling results in an effective magnetic field gradient within the scattering plan, exhibiting a spin-dependent scattering with different final momentum directions for spin-up and spin-down. In this case, the anomalous Hall resistivity linearly depends on the longitudinal resistivity, namely, $\rho_{xy} - \rho_{xx}$. Finally, the side jump, proposed by Berger [15], is due to a spin-dependent difference in electron acceleration and deceleration processes during the scattering, which effectively leads to a spin-dependent sideway displacement upon repeated scattering. In experiment, the side jump scattering mechanism gives the same scaling relation as the intrinsic contributions. By changing the temperature and the density of states at the Fermi surface, the power law relationship between ρ_{xy} and ρ_{xx} has been extensively studied experimentally in a large series of ferromagnetic materials [16], [20]–[22]. However, the scaling law of AHE is still lacking for ferrimagnetic materials, which includes two sets of magnetic lattices.

As a typical ferrimagnet, the magnetization of CoGd thin films can be controlled through modifying the alloy composition [23], [24]. When Gd contributes larger magnetization than that of Co, the magnetization of Co is antiparallel to that of CoGd thin films, and vice versa.

In this paper, we systematically investigated the magnetic and transport properties (Hall and longitudinal magnetoresistance) of ferrimagnetic Co₉₀Gd₁₀ by changing the thicknesses from 5 to 35 nm. We showed that the saturation magnetization of Co₉₀Gd₁₀ is enhanced by 14% when the temperature increases from 80 to 300 K. Surprisingly, the scaling law $\sigma_{xy} - \sigma_{xx}^n$ with $n = 2.3$ still holds regardless of thickness and temperature, in a good agreement with the universal scaling law reported in ferromagnetic materials.

II. EXPERIMENTAL DETAILS

The sample of Co₉₀Gd₁₀ was deposited on Si (100) at ambient temperature by magnetron sputtering system. The

Manuscript received March 20, 2015; accepted June 1, 2015. Date of publication June 10, 2015; date of current version October 22, 2015. Corresponding author: D. Z. Yang (email:yangdzh@lzu.edu.cn) and M. S. Si (email:sims@lzu.edu.cn).

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Digital Object Identifier 10.1109/TMAG.2015.2443112

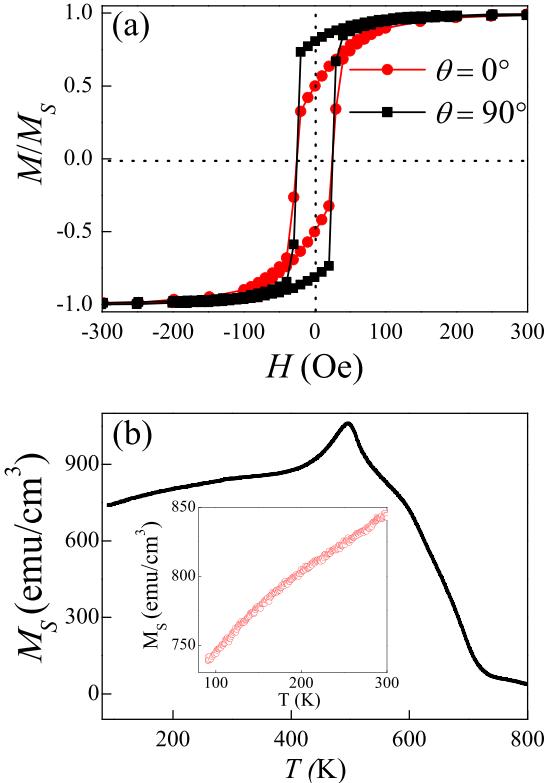


Fig. 1. (a) In-plane hysteresis loops of Co₉₀Gd₁₀ at RT with $\theta = 0$ and 90° , where θ is the in-plane angle between the easy axis and magnetic field. (b) Saturation magnetization of Co₉₀Gd₁₀ as a function of temperature. Inset: the temperature-dependent saturation magnetization zoomed-in view from 80 to 300 K.

compositions of Co₉₀Gd₁₀ layers were analyzed by energy dispersive X-ray spectroscope. Structural characterization was carried out by X-ray diffraction. Here, all Co₉₀Gd₁₀ films are amorphous (not shown for brevity). To analyze the magnetic properties, magnetization was measured by vibrating sample magnetometer. The transport properties of samples were measured by current source (Keithley 220) and voltagemeter (Keithley 2182).

III. RESULTS AND DISCUSSION

Fig. 1(a) shows the typical in-plane hysteresis loops for Co₉₀Gd₁₀ at room temperature (RT). The coercivity of Co₉₀Gd₁₀ is only about 27 Oe. For $\theta = 0$ and 90° , a weak anisotropy is observed due to the amorphous structure, where θ is the in-plane angle between the easy axis and magnetic field. Fig. 1(b) shows the saturation magnetization (M_S) of Co₉₀Gd₁₀ as a function of temperature. As expected, the M_S of Co₉₀Gd₁₀ can be tuned significantly by changing the temperature. When temperature increases from 80 to 300 K, the M_S of Co₉₀Gd₁₀ also significantly increases from 720 to 850 emu/cm³, as shown in the inset of Fig. 1(b). Remarkably, this temperature dependence of M_S of Co₉₀Gd₁₀ contrasts to that of ferromagnet (e.g., Co, Fe, and FeNi), in which M_S almost keeps constant in the temperature range from 80 to 300 K. The distinctive temperature behavior of Co₉₀Gd₁₀ can be explained by the difference between the antiparallel Co and Gd magnetic

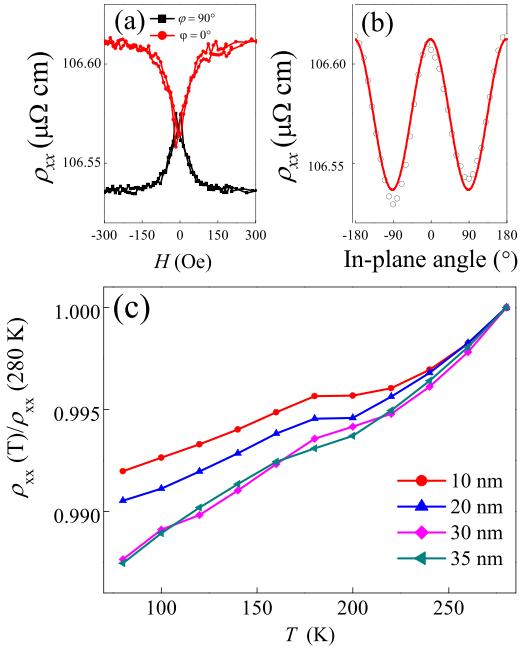


Fig. 2. (a) Longitudinal magnetoresistance of Co₉₀Gd₁₀ at RT for $\varphi = 0$ and 90° , where φ is the angle between the current and magnetic field in plane. (b) Angular dependence of magnetoresistance of Co₉₀Gd₁₀ at RT. (c) Temperature dependence of longitudinal resistance of Co₉₀Gd₁₀ with different thicknesses.

moments. As the temperature further increases and is close to 700 K, which reaches the Curie temperature of Co₉₀Gd₁₀, the MS of Co₉₀Gd₁₀ sharply declines to zero. One can also notice a peak at $T = 580$ K in Fig. 1(b), which might be caused by the crystallization in part of Co₉₀Gd₁₀.

Similar to ferromagnetic materials anisotropy magnetoresistance (AMR) of Co₉₀Gd₁₀ at $\varphi = 0$ and 90° is shown in Fig. 2(a), where φ is the angle between the current and magnetic field in plane. In order to further confirm the AMR of Co₉₀Gd₁₀, we measured the angular dependence of resistivity curves, where a specified magnetic field at 800 Oe is chosen to saturate the magnetization of Co₉₀Gd₁₀. It shows that the resistivity as a function of angle φ can be fitted well by $\cos^2(\varphi)$, as shown as the solid line in Fig. 2(b). The $\cos^2(\varphi)$ angular dependence is consistent with the AMR observed in the ferromagnetic materials due to the spin-orbit coupling. However, one can also find that the magnetoresistance (MR) ratio here is extremely low (only about 0.08%), due to the weak scattering of Gd magnetic moment, where the 4f magnetic moment of Gd atoms is far away the Fermi surface. The weak AMR effect of Co₉₀Gd₁₀ also indicates that the magnetic scattering almost has no contribution to the longitudinal resistivity of Co₉₀Gd₁₀. We noticed a difference for the ρ_{xx} at $\varphi = \pm 90^\circ$. This might be caused by the sample is tilted against the magnet. Fig. 2(c) shows the resistivity of Co₉₀Gd₁₀ as a function of temperature from 80 to 280 K. For all the thicknesses of Co₉₀Gd₁₀, the resistivity is largest at 280 K and gradually decays with the decreasing temperature. However, below 180 K, a linear decrease of resistivity is observed. The two types of temperature dependence indicate the competition scattering mechanisms. For the temperature range from 80 to 180 K, the

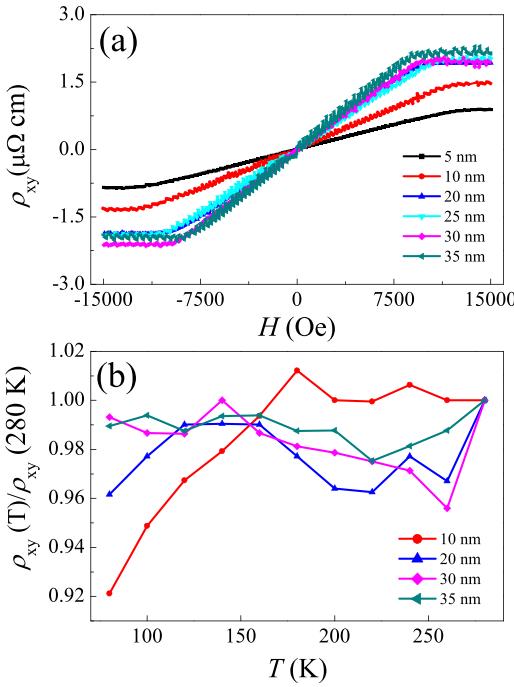


Fig. 3. (a) Hall resistance of Co₉₀Gd₁₀ as a function of the magnetic field measured at RT with different thicknesses. (b) Temperature dependent of Hall resistance for different thicknesses Co₉₀Gd₁₀.

magnetic scattering dominants the longitudinal resistivity. For the temperature above 180 K, the impurity scattering plays an important role in resistivity. Interestingly, the longitudinal resistivity is almost independent with the temperature for all thicknesses, despite the magnetization of Co₉₀Gd₁₀ can be tuned 14%.

The transverse Hall resistivities for all thicknesses from 5 to 35 nm were measured at temperatures with range from 80 to 300 K with the magnetic field applied perpendicular to the sample plane. Fig. 3(a) shows the magnetic field dependence of the Hall resistivity at RT. The magnetization is saturated until the magnetic field reaches about 1.0 T. Above the saturation field, a weak linear increase of Hall resistivity is caused by the ordinary Hall effect. One can also find that the transverse Hall resistivity strongly depends on the thicknesses. It obviously increases with the increasing thicknesses. Remarkably, the Hall resistivity of Co₉₀Gd₁₀ increases upon increasing the thickness, whereas in ferromagnetic materials, the Hall resistivity decreases with the thickness [19]. Fig. 3(c) shows the Hall resistivities of Co₉₀Gd₁₀ as a function of temperature from 80 to 280 K for different thicknesses. Surprisingly, similar to the longitudinal resistivity, it can be seen that these values of anomalous Hall resistivity are weakly related to temperatures.

The longitudinal conductivity σ_{xx} and the transverse anomalous conductivity σ_{xy} were estimated through the relations $\sigma_{xx} = \rho_{xx}/(\rho_{xx}^2 + \rho_{xy}^2)$ and $\sigma_{xy} = \rho_{xy}/(\rho_{xx}^2 + \rho_{xy}^2)$, respectively, which can be approximated as $\sigma_{xx} \cong 1/\rho_{xx}$ and $\sigma_{xy} \cong \rho_{xy}/\rho_{xx}^2$ in the different Co₉₀Gd₁₀ thicknesses because the absolute value of $\rho_{xy} \ll \rho_{xx}$. The relationship between σ_{xx} and σ_{xy} of Co₉₀Gd₁₀ has been concluded in Fig. 4. Due

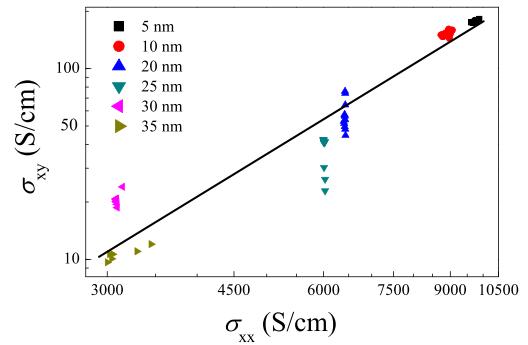


Fig. 4. Anomalous Hall conductivity σ_{xy} as a function of longitudinal conductivity σ_{xx} for Co₉₀Gd₁₀. The solid lines are fit to $\sigma_{xy} - \sigma_{xx}^{2.3}$.

to $\sigma_{xx} < 10^4$ S/cm, our results can be considered to belong in the dirty limit region, where the relation $\sigma_{xy} - \sigma_{xx}^n$ with $n \cong 2$ is reported in ferromagnet [16]. By fitting all our data, the value of $n = 2.3$ is obtained, which is also consistent with the results in ferromagnetic materials.

IV. CONCLUSION

In conclusion, we studied the scaling law of the ferrimagnet Co₉₀Gd₁₀, by setting various thicknesses and temperatures as control parameters to tune both σ_{xx} and σ_{xy} . The scaling law $\sigma_{xy} - \sigma_{xx}^n$ with $n = 2.3$ surprisingly holds in reasonable agreement with theory, which further confirms the universality of the scaling law. Our results also indicate that the AHE in Co₉₀Gd₁₀ is belong in the dirty limit region due to the doping.

ACKNOWLEDGMENT

This work was supported in part by the National Basic Research Program of China under Grant 2012CB933101, in part by the NSFC of China under grant 51372107, grant 51201081, and grant 51202102, in part by the Gansu Province Science and Technology Funding under grant 1208RJYA008, in part by the Fundamental Research Funds for the Central Universities under Grant lzu-jbky-2014-16 and Grant lzu-jbky-2013-23, and in part by the Program for Changjiang Scholars and Innovative Research Team in University under Grant IRT1251.

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